Hole mobility enhancement in In_{0.41}Ga_{0.59}Sb quantum-well field-effect transistors

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The impact of $\langle 110 \rangle$ uniaxial strain on the characteristics of p-channel In_{0.41}Ga_{0.59}Sb quantum-well field-effect transistors (QW-FETs) is studied through chip-bending experiments. Uniaxial strain is found to affect the linear-regime drain current and the threshold voltage of the FET through the modulation of the hole mobility of the two-dimensional hole gas (2DHG) in the QW-FET. The piezoresistance coefficients of the 2DHG have been determined to be $\pi^{\parallel}_{\langle 110 \rangle} = 1.17 \times 10^{-10} \text{ cm}^2/\text{dyn}$ and $\pi^{\perp}_{\langle 110 \rangle} = -1.9 \times 10^{-11} \text{ cm}^2/\text{dyn}$. The value of $\pi^{\parallel}_{\langle 110 \rangle}$ is 1.5 times that of holes in Si metal-oxide-semiconductor (MOS) field-effect transistors and establishes InGaSb as a promising material system for a future III-V complementary MOS (CMOS) technology. © 2011 American Institute of Physics. [doi:10.1063/1.3552963]

In recent times, there has been great interest in exploring the substitution of the Si channel in scaled logic field-effect transistors (FETs) with a III-V compound semiconductor.¹ In this quest for a III-V complementary logic technology, realizing a high performance p-channel III-V FET remains a great challenge. One of the reasons lies in the enormous difference between the electron and hole transport properties in III-V's. The hole mobility in most III-V's is at best of the same order as in Si. A notable exception is InGaSb which exhibits a high Hall mobility of ~1500 cm²/V s (Ref. 2) and has attracted considerable interest for future logic applications.³ Recently, p-channel InGaSb metal-oxidesemiconductor FETs (MOSFETs) with a high-quality dielectric interface have been demonstrated.⁴

A feasible approach to increase the hole mobility in the channel of a FET is to introduce strain. In fact, processinduced strain has become a crucial component of mainstream Si logic technology since the 90 nm node.⁵ Specifically, $\langle 110 \rangle$ uniaxial strain has been favored by industry because it brings a larger mobility enhancement along the strain direction than biaxial strain, especially under high surface electric field.⁶ In III-V systems, however, only a few reports^{4,7,8} exist that explore hole mobility enhancement through the introduction of $\langle 110 \rangle$ strain.

In this work, we carry out an experimental study of uniaxial strain effects on the hole mobility of twodimensional hole gas (2DHG) in $In_{0.41}Ga_{0.59}Sb$ quantumwell field-effect transistors (QW-FETs), which is an excellent model system to study physics of relevance to future logic p-type MOSFET based on this material. Compared with a notable recent work on $In_{0.35}Ga_{0.65}Sb$ MOSFETs with Al_2O_3 gate dielectric,⁴ the structure studied in our work provides an epitaxy-quality heterostructure interface and the highest mobility in this material system to date.

Figure 1 shows the cross section of the InGaSb QW-FET used in this study. The fabrication process has been reported elsewhere.² This heterostructure is characterized by a Hall

mobility of 1500 cm²/V s and a sheet hole density (p_s) of 6.6×10^{11} cm⁻². The channel is under 2.1% biaxial compressive strain, which contributes to the high hole mobility.⁹ The fabricated devices have a 0.2 μ m gate length. Other electrical characteristics of these devices are given in Ref. 10.

 $\langle 110 \rangle$ uniaxial strain was introduced to the QW-FET by a chip-bending apparatus.¹¹ Holes flow in the channel of these devices along the [110] direction. A maximum of 0.08% strain parallel and perpendicular to the channel was sequentially applied to the same FET by rotating the chip by 90° . The stress is calculated using an estimated $\langle 110 \rangle$ Young's modulus (78 GPa) (Ref. 12) for the $In_{0.41}Ga_{0.59}Sb$. The strain at the chip surface is known within $\sim 3.5\%$. Changes in the device characteristics with strain were found to be recoverable. To minimize heating effects and parasitic Ohmic drops, the transfer characteristics at low $V_{\rm DS}$ (-50 mV) were measured. We focused on two figures of merit: the threshold voltage ($V_{\rm T}$, extracted at $I_{\rm D}$ =0.1 mA/mm) was initially used as a proxy to study device electrostatics, while the linearregime drain current (I_{Dlin} , extracted at $V_{\text{GS}} - V_{\text{T}} = -0.2$ V) was used to study the 2DHG channel mobility. The total channel resistance of our FET can be written as $R_{\rm ch}=2R_{\rm c}$ $+2R_{\text{ext}}+R_{\text{int}}$, where R_{c} is the contact resistance between Ohmic metal and the 2DHG underneath it, R_{ext} is the resistance of the ungated semiconductor portion between the gate and the source/drain (S/D) contact metals, and R_{int} is the



FIG. 1. (Left) The cross section of the $In_{0.41}Ga_{0.59}Sb$ QW-FET. (Right) Measured transfer characteristics as σ_{\parallel} changes.

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FIG. 2. (Color online) Relative change of linear-regime drain current at $V_{\rm GS}-V_{\rm T}$ =-0.2 V as a function of $\langle 110 \rangle$ stress. Solid lines are linear fittings to the data.

resistance of the intrinsic region under gate. $2R_c$ measured by the transmission line method is ~4 Ω mm. This is <6% of R_{tot} (~68 Ω mm) at the bias of the I_{Dlin} extraction. R_{ext} is considered to be governed by similar strain effects as to R_{int} , except for a fixed 2DHG concentration of 6.6×10^{11} cm⁻². Therefore, any impact from a fixed parasitic resistance arises only from $2R_c$, which can be safely neglected.

Figure 1 shows a representative example of the change in the transfer characteristics of an InGaSb QW-FET with stress applied along the channel direction (σ_{\parallel}). A significant increase in I_{Dlin} with compressive stress ($\sigma < 0$) was observed.

Figure 2 summarizes the change of I_{Dlin} at $V_{\text{GS}}-V_{\text{T}}=-0.2$ V measured with both stress parallel (σ_{\parallel}) and perpendicular (σ_{\perp}) to the channel. Significant anisotropic effects are seen: the magnitude of ΔI_{Dlin} under σ_{\parallel} is ~6 times higher than under σ_{\perp} ; the signs are also opposite for ΔI_{Dlin} with σ_{\parallel} and σ_{\perp} . For high gate overdrive, $-0.2 \text{ V} > V_{\text{GS}} - V_{\text{T}} > -0.4$ V (the maximum in this study), $\Delta I_{\text{Dlin}}/I_{\text{Dlin}}$ is found to be independent of V_{GS} . This is because at high gate overdrive, the intrinsic resistance (R_{int}) becomes small compared with that of the S/D regions (R_{ext}). Therefore, $\Delta I_{\text{Dlin}}/I_{\text{Dlin}}$ under high gate overdrive is dominated by $\Delta R_{\text{ext}}/R_{\text{ext}}$, and is independent of V_{GS} .

Attributing the change in I_{Dlin} to a change of μ_{h} requires some caution, because I_{Dlin} depends on p_s as well. Our previous study¹¹ showed that $\langle 110 \rangle$ uniaxial stress changes the two-dimensional (2D) carrier concentration in InGaAs QW-FETs through the piezoelectric effect by changing $V_{\rm T}$ and the gate capacitance ($C_{\rm G}$). Shifts in $V_{\rm T}$ are also seen in the current devices (Fig. 3). However, changes of p_s due to the piezoelectric effect in the current InGaSb QW-FET are estimated to be negligible. One-dimensional Schrödinger-Poisson simulations show that Δp_s or ΔC_G due to the piezoelectric effect is 25 times smaller than the observed ΔI_{Dlin} . The reason is the tight confinement of 2DHG by the extremely thin quantum well (7.5 nm) and the small piezoelectric constants of the materials involved.¹³ Therefore, we can conclude that the observed ΔI_{Dlin} is induced by $\Delta \mu_{\text{h}}$. This requires a fresh explanation for the change in $V_{\rm T}$ seen in Fig. 3. We postulate that this is also caused by $\Delta \mu_{\rm h}$.

Since we extract $V_{\rm T}$ at a constant current in the subthreshold regime, anything that affects the subthreshold current ($I_{\rm Dsub}$) can propagate into an apparent change in $V_{\rm T}$.



FIG. 3. (Color online) Change of measured or apparent threshold voltage as a function of $\langle 110 \rangle$ stress. The solid lines represent $\Delta V_{\rm T}^{\rm app}$ projected from $\Delta \mu_{\rm h}$ according to Eq. (2).

Theoretically, carrier transport in the subthreshold regime of a FET follows a diffusion process.¹⁴ Therefore, I_{Dsub} depends on μ_{h} through the Einstein relation and the difference in p_{s} at the source and drain edges of the gate. Approximately, p_{s} depends linearly on the effective 2D density of states (DOS), and exponentially on V_{GS} .¹⁵ Therefore, for $V_{\text{GS}} - V_{\text{T}} \gg kT/q$,

$$I_{\rm Dsub} \propto \mu_{\rm h} \exp \frac{-q(V_{\rm GS} - V_{\rm T})}{nkT},\tag{1}$$

where *n* is the ideality factor, *k* is the Boltzmann constant, *T* is the temperature, and $V_{\rm T}$ is the threshold voltage which we define as the condition in which the Fermi level lines up with the top of the first subband in the channel.

This model suggests that I_{Dsub} is linearly proportional to μ_{h} . The proportionality constant in Eq. (1) contains the 2D DOS. $8 \times 8 \ k \cdot p$ simulations suggest that changes to the DOS due to strain are negligible [Fig. 4(c)] for our level of stress and only lead to $|\Delta V_{\text{T}}| < 0.5 \text{ mV}$. The parameters used in these simulations are according to Ref. 16. The effects of built-in biaxial strain and quantization are included in the



FIG. 4. In-plane dispersion relationship of the higher lying band in the In_{0.41}Ga_{0.59}Sb QW under (a) 62 MPa tensile stress and (b) 62 MPa compressive stress. (c) Density of states in In_{0.41}Ga_{0.59}Sb valence band for different values of stress. The change in density of states only leads to a marginal $\Delta V_{\rm T}$ at the current level of stress.

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simulations. In consequence, the change in apparent $V_{\rm T}$ measured at constant subthreshold current $(\Delta V_{\rm T}^{\rm app})$ is indeed a shift of $V_{\rm GS}$ in the following way:

$$\Delta V_{\rm GS} = \Delta V_{\rm T}^{\rm app} = \Delta V_{\rm T}^{\rm elec} + \frac{nkT}{q} \ln \left(1 + \frac{\Delta \mu_{\rm h}}{\mu_{\rm h}} \right). \tag{2}$$

 $\Delta V_{\rm GS}$ consists of a term that captures the change induced by device electrostatics ($\Delta V_{\rm T}^{\rm elec}$) plus another term that includes mobility changes. $\Delta V_{\rm T}^{\rm elec}$ is affected by the piezoelectric effect and Schottky barrier height ($\phi_{\rm B}$) changes.¹¹ However, under the conditions of the present study, as mentioned above, the impact of the piezoelectric effect and Schottky barrier height changes on $\Delta V_{\rm T}^{\rm elec}$ are <0.5 and <0.6 mV, respectively. Therefore, we can conclude that $\Delta V_{\rm T}^{\rm app}$ is dominated by $\Delta \mu_{\rm h}$ induced change.

As shown in Fig. 3, calculations of $\Delta V_{\rm T}^{\rm app}$ using this relation broadly agree with our experiments. The solid lines show the projected $\Delta V_{\rm T}^{\rm app}$ as a result of $\Delta \mu_{\rm h}$ extracted from $\Delta I_{\rm Dlin}$. The projected $\Delta V_{\rm T}^{\rm app}$ matches relatively well with the apparent $\Delta V_{\rm T}$ extracted from the subthreshold regime. The residual gap in Fig. 3 between the model and the data may be attributed to a larger $\Delta \mu_{\rm h} / \Delta \sigma$ in the subthreshold regime than in the linear regime. This effect is akin to the decrease in the piezoresistance coefficients in p-type Si with increased carrier concentration.¹⁷

The piezoresistance coefficients $(\pi = -\Delta \mu / \mu \sigma)$ of the In_{0.41}Ga_{0.59}Sb 2DHG parallel and perpendicular to $\langle 110 \rangle$ uniaxial stress can then be calculated from the data in Fig. 2. They are found to be $\pi^{\parallel}_{\langle 110 \rangle} = 1.17 \times 10^{-10} \text{ cm}^2/\text{dyn}$ and $\pi^{\perp}_{\langle 110 \rangle} = -1.9 \times 10^{-11} \text{ cm}^2/\text{dyn}$. Compared with $\pi^{\parallel}_{\langle 110 \rangle}$ for Si pMOS at a similar hole concentration $(6.6 \times 10^{11} \text{ cm}^{-2})$,^{18,19} $\pi^{\parallel}_{\langle 110 \rangle}$ of the In_{0.41}Ga_{0.59}Sb 2DHG is 1.5 times higher. This value is also 1.4 times higher than the $\pi^{\parallel}_{\langle 110 \rangle}$ found in an In_{0.35}Ga_{0.65}Sb MOSFET in Ref. 4. Nevertheless, $\pi^{\parallel}_{\langle 110 \rangle}$ in Ref. 4 was probably measured at higher p_s (>10¹² cm⁻²), which might decrease its value.

The anisotropic behavior of the piezoresistance coefficients of $In_{0.41}Ga_{0.59}Sb$ qualitatively agrees with the trend seen in Si. This is partly due to the anisotropic response of the valence band to $\langle 110 \rangle$ uniaxial strain as seen in our $k \cdot p$ simulations (Fig. 4). More sophisticated calculations²⁰ are needed to theoretically quantify the change in hole mobility. A final remark is that the piezoresistance coefficients are measured for $In_{0.41}Ga_{0.59}Sb$ channel with 2.1% compressive biaxial strain and quantum confinement. These coefficients could be different under other built-in biaxial strain or confinement conditions.

In summary, we have experimentally studied the impact of $\langle 110 \rangle$ uniaxial strain on the electrical characteristics on p-channel In_{0.41}Ga_{0.59}Sb QW-FETs. We have found that $\langle 110 \rangle$ uniaxial strain can significantly enhance the hole mobility in these devices. The conclusion is confirmed by analysis consisting of Schrödinger–Poisson simulations, $k \cdot p$ simulations, and our model which reveals the relation between observed change in drain current and threshold voltage. The piezoresistance coefficients are determined to be $\pi^{\parallel}_{\langle 110\rangle} = 1.17 \times 10^{-10} \text{ cm}^2/\text{dyn}$ and $\pi^{\perp}_{\langle 110\rangle} = -1.9 \times 10^{-11} \text{ cm}^2/\text{dyn}$. $\pi^{\parallel}_{\langle 110\rangle}$ is 1.5 times more than in Si. Therefore, process-induced uniaxial strain should be valuable for enhancing the performance of p-channel In_{0.41}Ga_{0.51}Sb QW-FETs. When coupled with its high hole mobility, In_{0.41}Ga_{0.51}Sb emerges as a promising channel material for future high performance p-type logic FETs.

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